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Short-Circuit Diffusion in Recrystallizing Microstructure: Diffusion Properties of a Recrystallization Front

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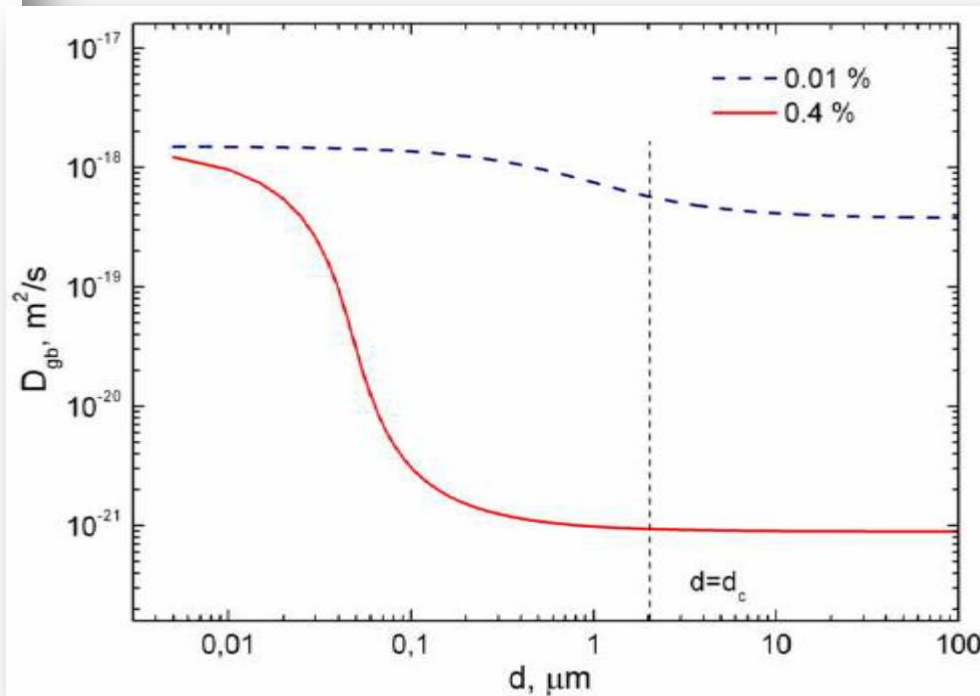
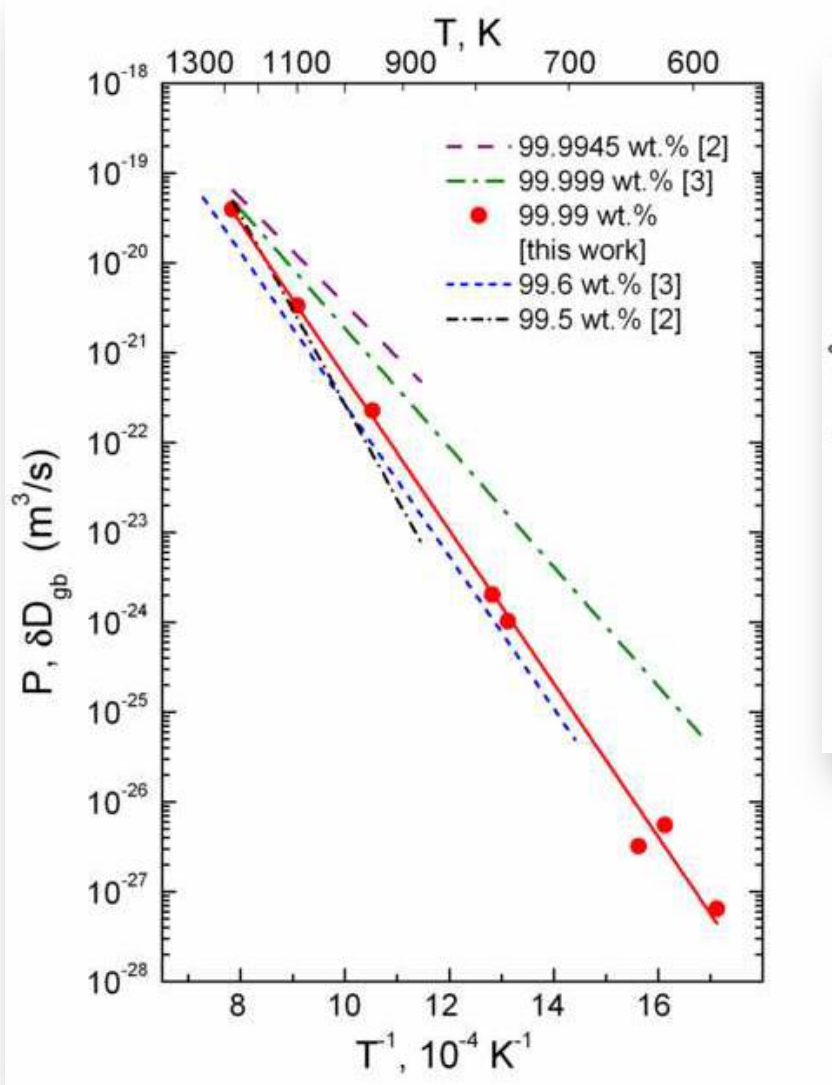
- Types of interfaces in recrystallizing material
- “Recrystallization front”
- Diffusion properties of stationary boundaries
- Diffusion enhancement in moving boundary

GB diffusion in coarse-grained Ni



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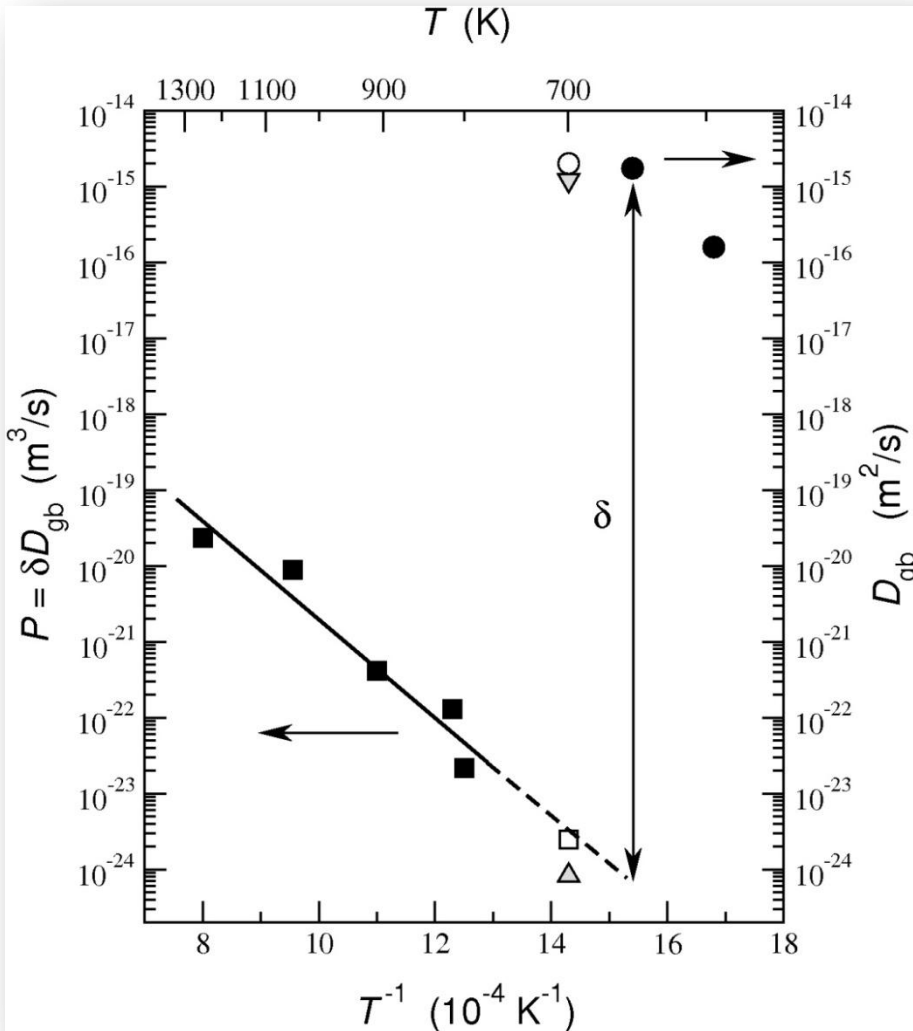
Interfaces in SPD materials



Effect of grain size and purity on GB diffusion

Prokoshkina, Esin, Wilde, S.D., Acta Mater (2013)

Grain boundary width



Material	Tracer	δ (nm)
Ni (5N, 4N)	^{63}Ni	0.54 ± 0.1
NiO	^{63}Ni	0.7^*
Ag	$^{110\text{m}}\text{Ag}$	$0.43 \pm 0.27^{**}$
Ag	$^{110\text{m}}\text{Ag}$	0.5^{***}
γ -FeNi	^{59}Fe	0.5
γ -FeNi	^{63}Ni	0.55 ± 0.43
α -Fe	^{59}Fe	0.5^{****}
α -Ti	^{44}Ti , ^{57}Co	0.5
Ag	$^{110\text{m}}\text{Ag}$	0.5

* Atkinson & Taylor, 1981

** Sommer & Herzig, 1992

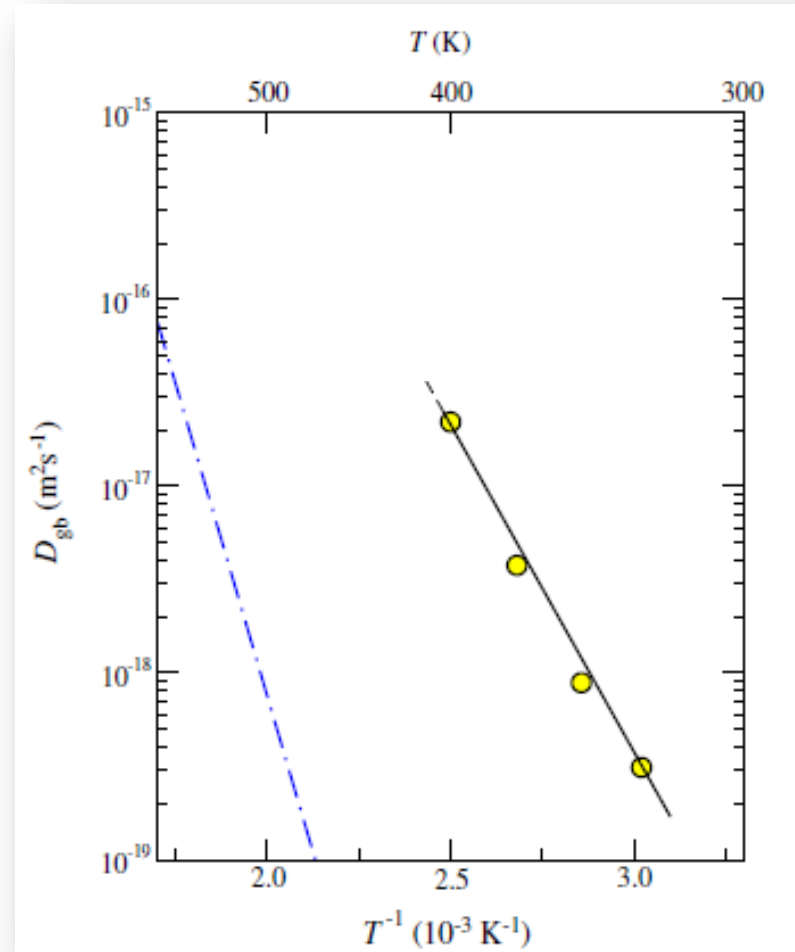
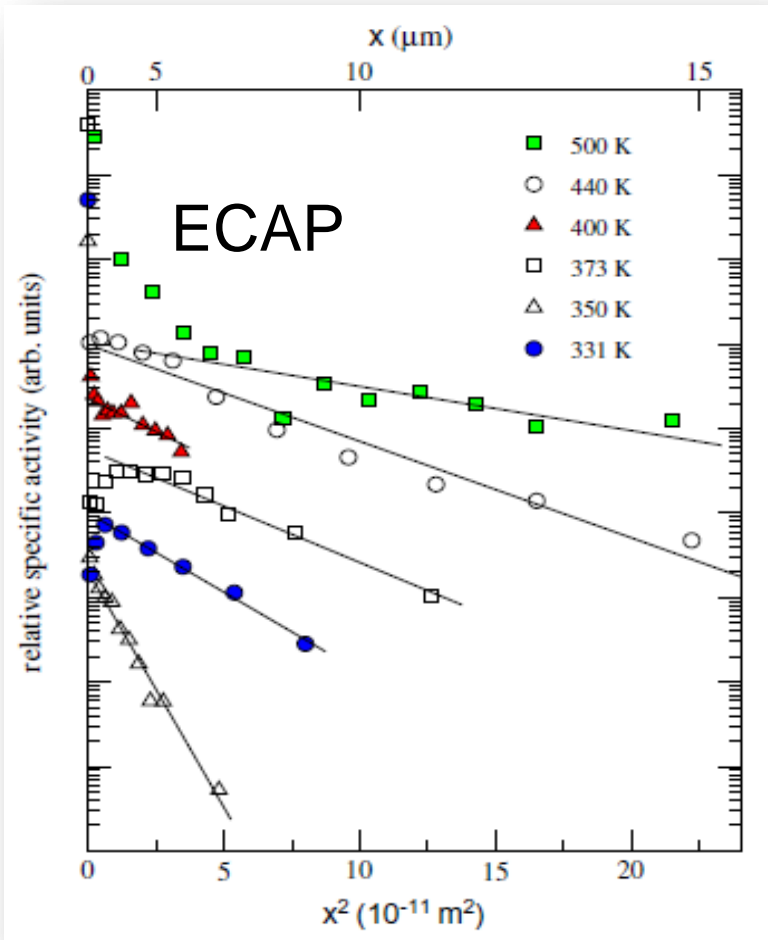
*** Bernardini et al, 1992

**** Iijima et al., 2007

Divinski et al, 2003, 2004, 2009, 2010, 2013

- Dislocations
- Dislocation walls, low-angle grain boundaries
- High-angle grain boundaries
 - General (random)
 - Special (sigma), twin boundaries
- **Deformation-modified interfaces**
(higher energy, higher coefficient of self-diffusion, strain fields)

GB diffusion in SPD Ni (99.6 wt.%)



Deformation-induced enhancement of GB diffusion in SPD Ni

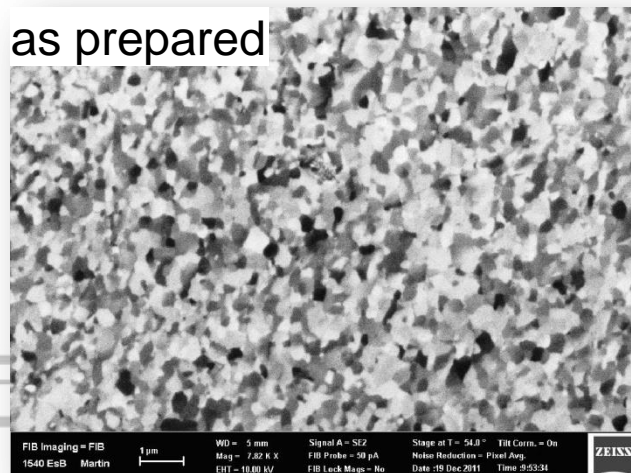
Divinski, Reglitz, Wilde, Acta Mater (2011)



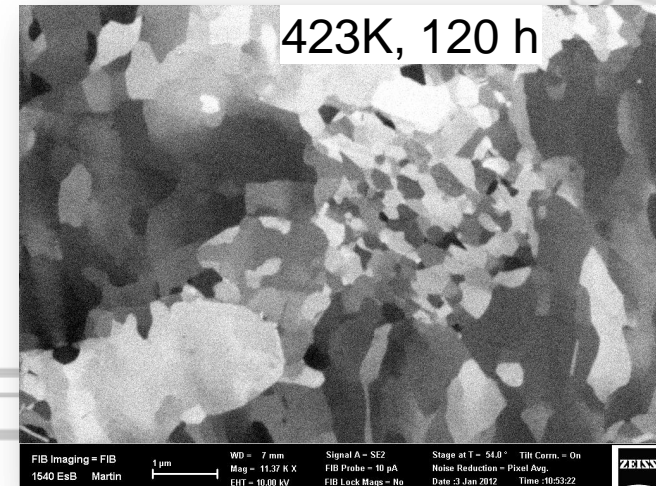
- Dislocations
- Dislocation walls, low-angle grain boundaries
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 - General (random)
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heterogenous recrystallization

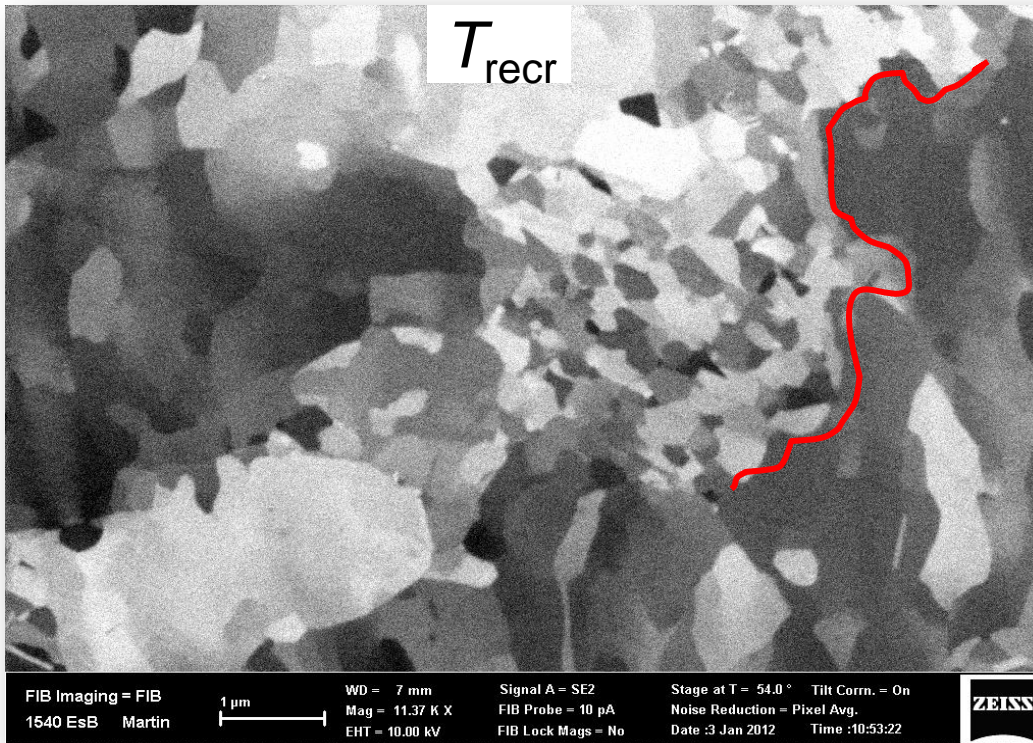
- Boundaries between newly appeared grains
- Boundaries between new and old (deformed) grains



Deformed Ni



Two strategies



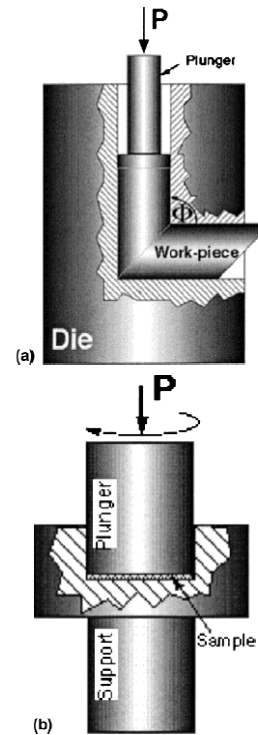
Diffusion experiments
at $T < T_{\text{recr}}$
⇒ stationary
recrystallization front

Diffusion experiments at $T = T_{\text{recr}}$
⇒ mobile recrystallization front



Short-circuit diffusion in deformed materials

- Rolling to 80%
- High-pressure torsion
- Radiotracer diffusion as a unique and sensitive probe for detecting changes of the interface structure

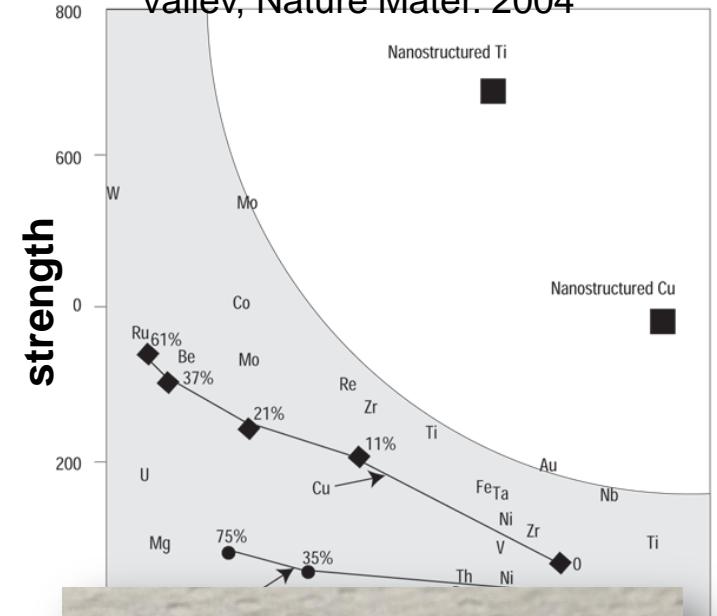


R.Z. Valiev, et al.,
J. Mater. Res., 2002

Interfaces in SPD materials

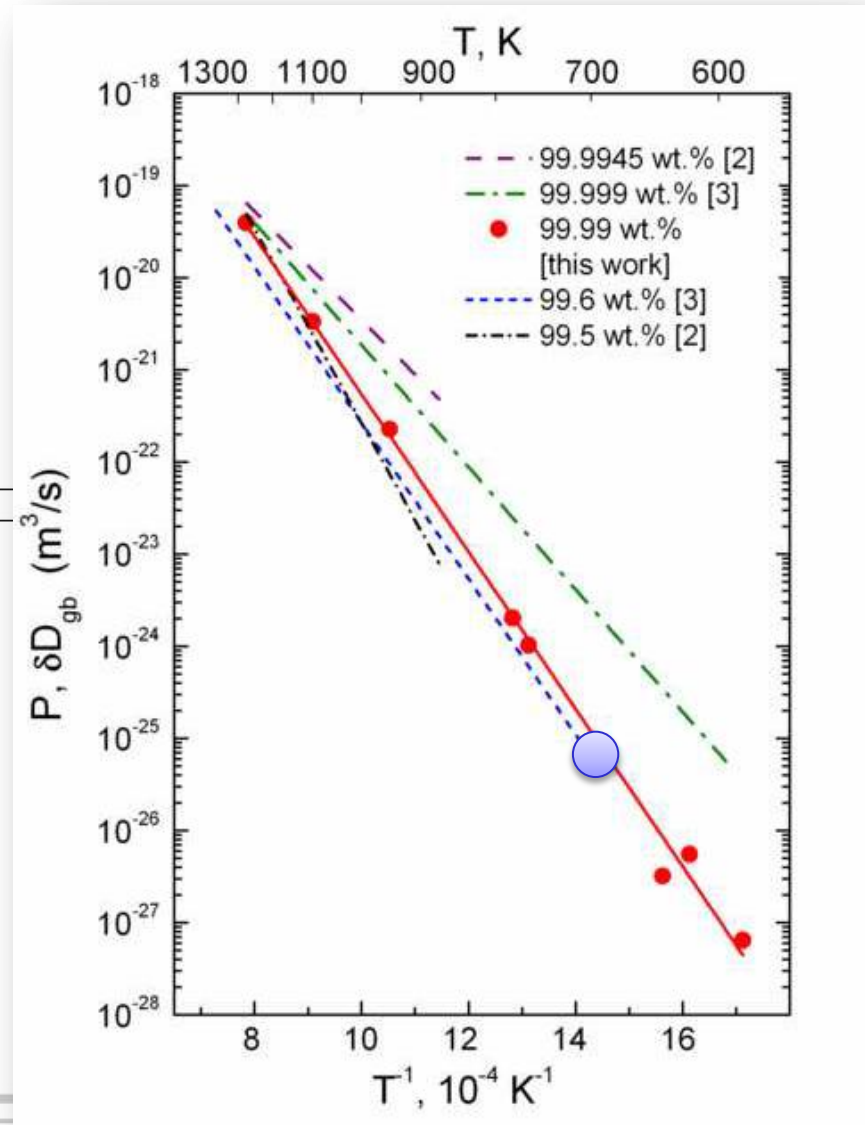
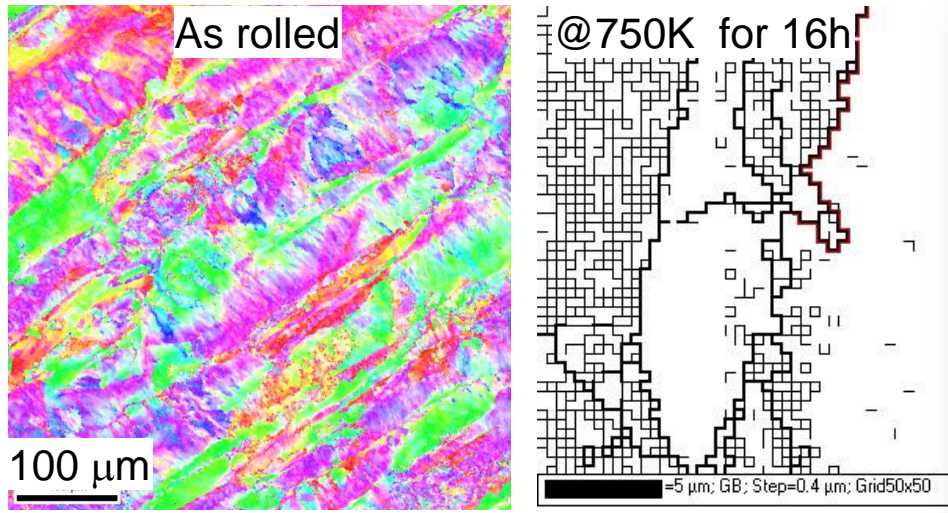
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Valiev, Nature Mater. 2004





Partially recrystallized rolled Ni (99.6wt%)

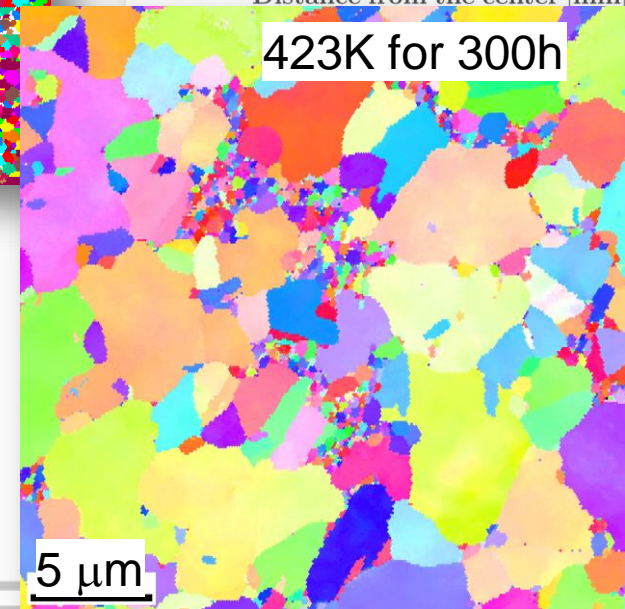
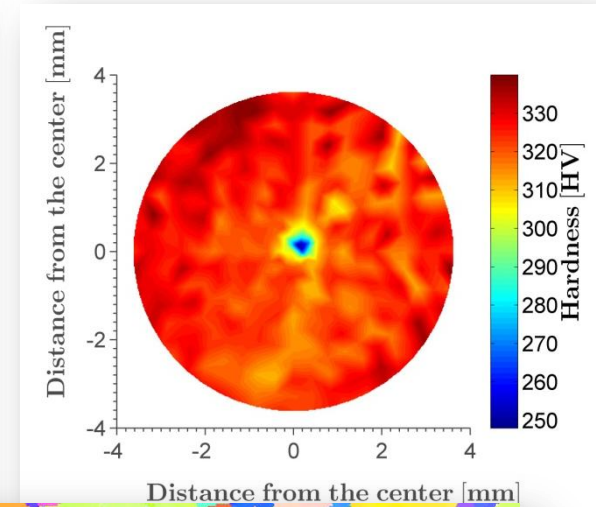
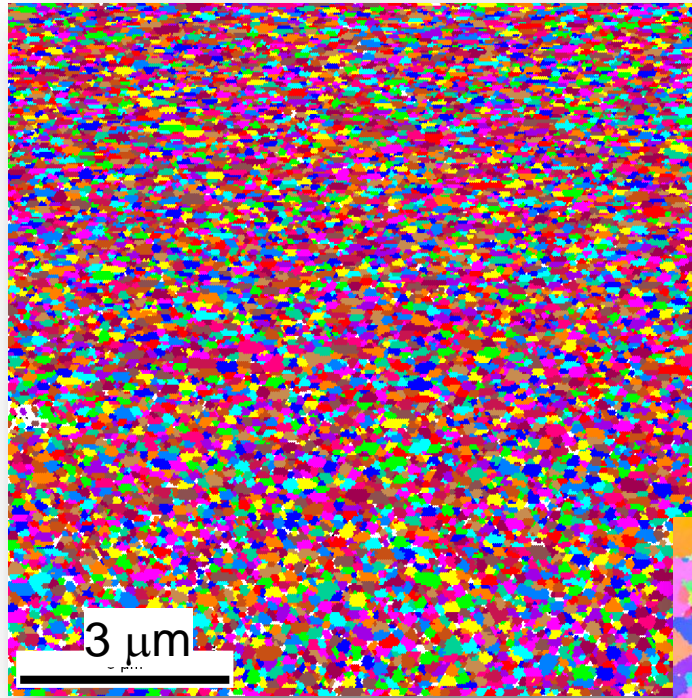
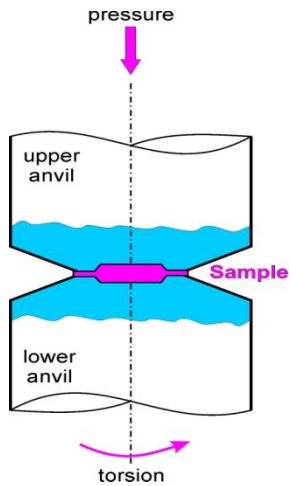


- no measurable diffusion enhancement at $T \leq 700 \text{ K}$
- **Kinetic properties of stable recrystallization front correspond to those of general high-angle GBs**



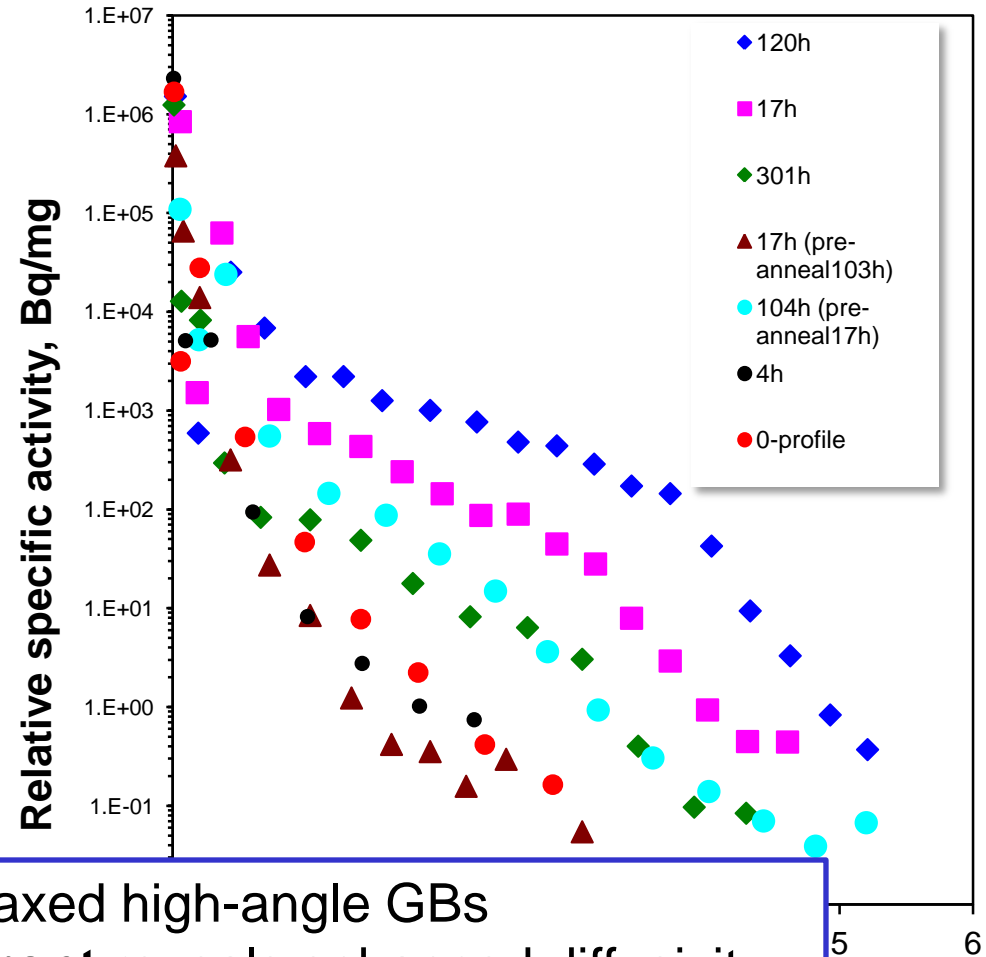
Diffusion and recrystallization in HPT Ni (99.99wt.%)

Interfaces in SPD materials

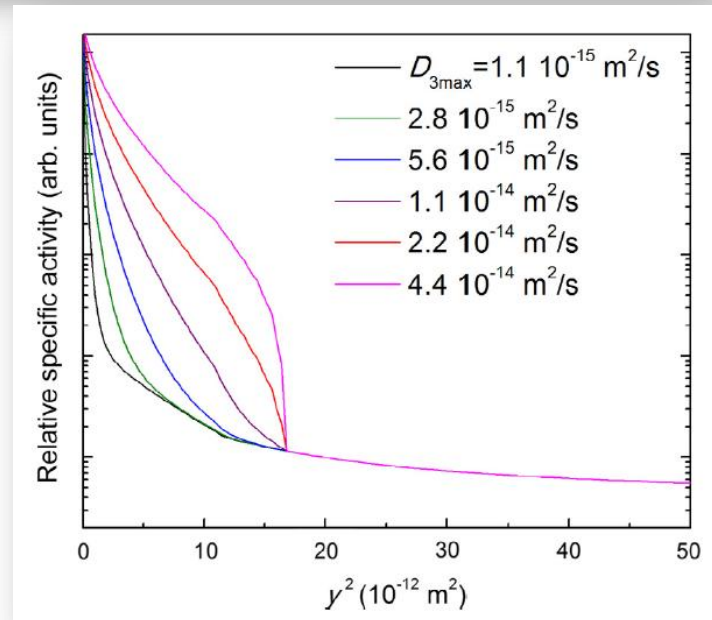
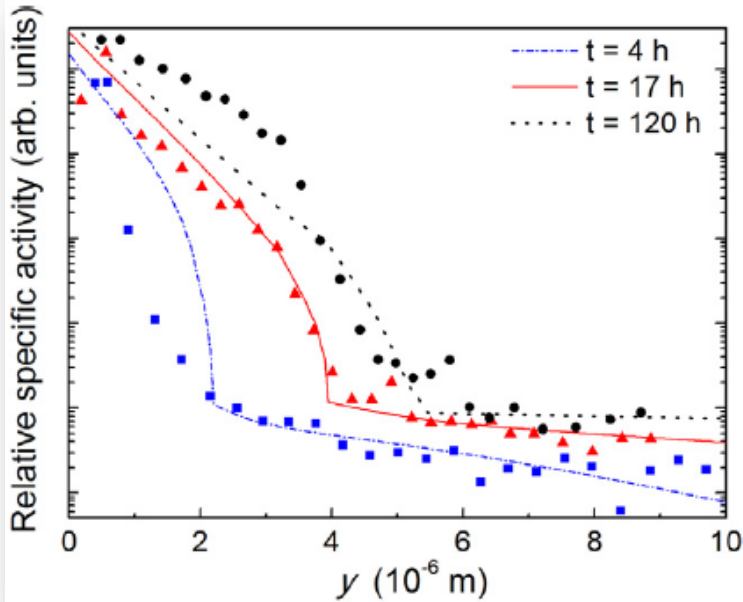
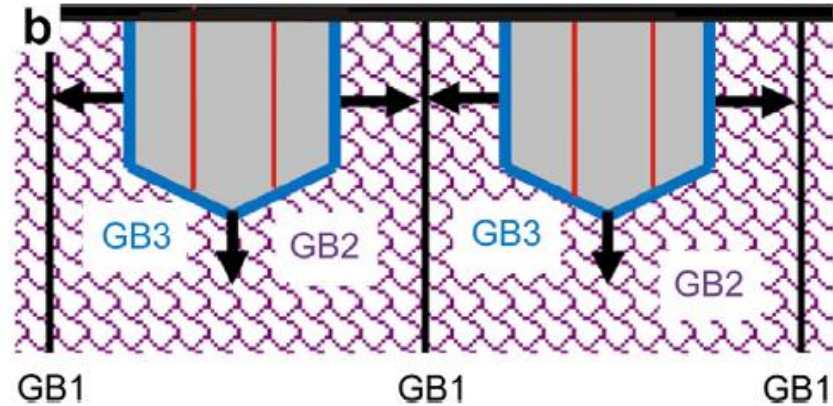
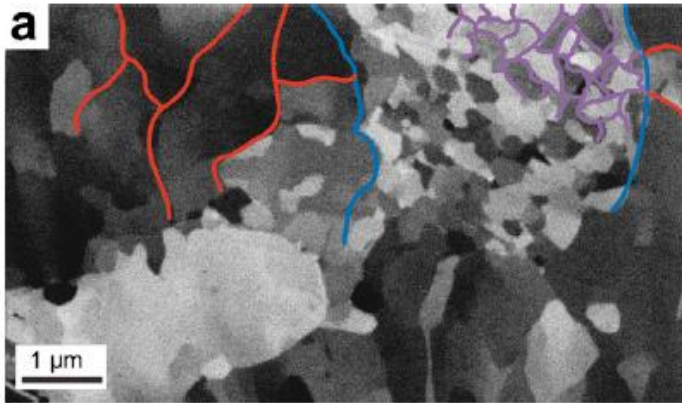


- 4N Ni is less stable against recrystallization / grain growth as 2N6 Ni
- Pronounced recrystallization at 423 K

- Diffusion component associated with deformation-induced state of GBs
- Diffusion component associated with new grain boundaries:
- Diffusion depths grow first
0h = 4h < 17h < 120h
and then decreases
120h > 301h
- Characteristic “kink” in the profiles measured for 17 and 120h
- No measurable diffusion after 104h pre-annelaing



- New GBs correspond to relaxed high-angle GBs
- **moving recrystallization front** reveals enhanced diffusivity
- stationary recrystallization front does not



Prokoshkina, et al, Acta Mater (2014)



Nickel:

- Deformation-induced diffusion enhancement
- Recrystallization:
 - intricate interplay of grain growth and purity on diffusivity of new GBs
 - Stationary recrystallization front reveals no measurable diffusion enhancement
 - Mobile recrystallization front represents a fast-diffusion path

Copper

- Enhanced GB diffusion in fully recrystallized microstructure at $T < 500\text{K}$
- „normal“ GB diffusion in microstructures recrystallized at $T > 600\text{K}$

Silver

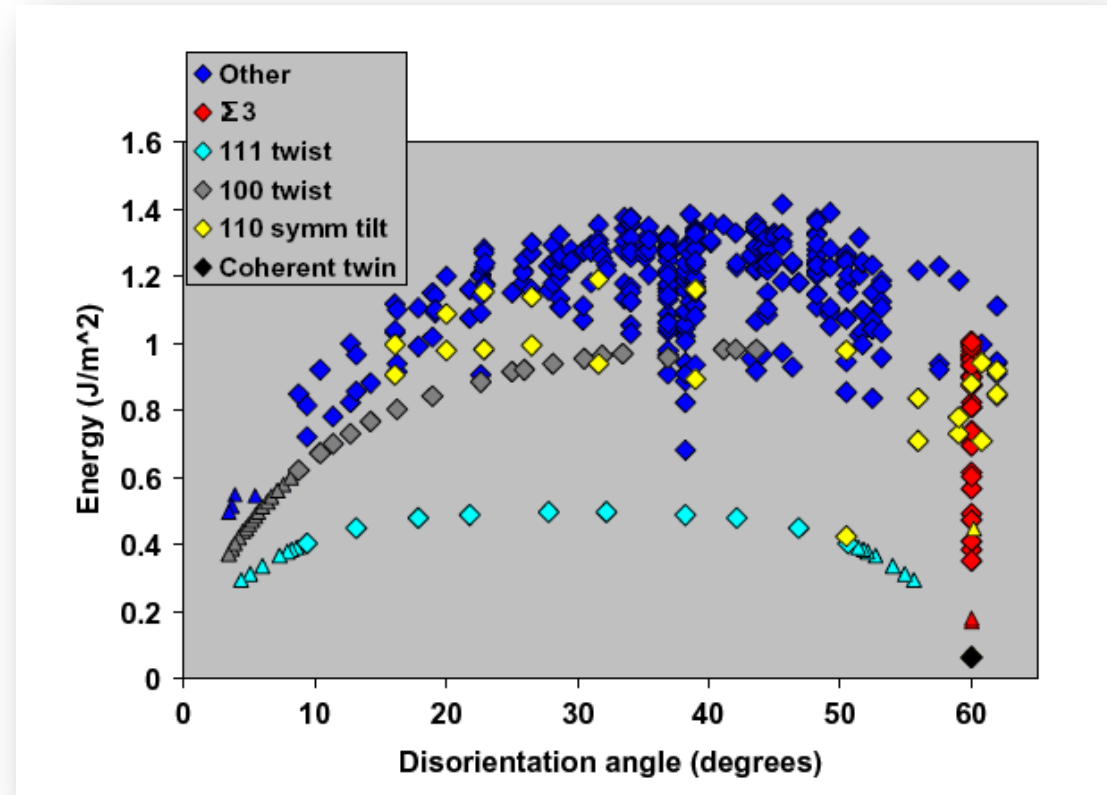
- Enhanced GB diffusion in dynamically recrystallized microstructure
- „Normal“ GB diffusion after recrystallization at $T > 500\text{K}$

Effect of free volume redistribution/annihilation

⇒ formation of voids and interconnected porosity



- Many „high energy” grain boundaries
- „constrained” recrystallisation at low temperature creates a high fraction of „high energy” grain boundaries



E. A. Holm et al., Acta Mater., 57, 3694-3703 (2009)

⇒ Ostwald step rule