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# 3D mapping of polymerization shrinkage using X-ray micro-computed tomography to predict microleakage\*

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#### ABSTRACT

Objectives. The objectives of this study were to (1) demonstrate X-ray micro-computed tomography ( $\mu$ CT) as a viable method for determining the polymerization shrinkage and microleakage on the same sample accurately and non-destructively, and (2) investigate the effect of sample geometry (e.g., C-factor and volume) on polymerization shrinkage and microleakage.

Methods. Composites placed in a series of model cavities of controlled C-factors and volumes were imaged using  $\mu CT$  to determine their precise location and volume before and after photopolymerization. Shrinkage was calculated by comparing the volume of composites before and after polymerization and leakage was predicted based on gap formation between composites and cavity walls as a function of position. Dye penetration experiments were used to validate  $\mu CT$  results.

Results. The degree of conversion (DC) of composites measured using FTIR microspectroscopy in reflectance mode was nearly identical for composites filled in all model cavity geometries. The shrinkage of composites calculated based on  $\mu$ CT results was statistically identical regardless of sample geometry. Microleakage, on the other hand, was highly dependent on the C-factor as well as the composite volume, with higher C-factors and larger volumes leading to a greater probability of microleakage. Spatial distribution of microleakage determined by  $\mu$ CT agreed well with results determined by dye penetration.

Significance. µCT has proven to be a powerful technique in quantifying polymerization shrinkage and corresponding microleakage for clinically relevant cavity geometries.

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## 1. Introduction

Polymeric dental composites are increasingly used as restorative materials because of their good aesthetic and mechanical properties; however, their long-term performance remains less than optimal, frequently resulting in costly replacement procedures. Over 50% of replacement procedures have

been attributed to the formation of recurrent (secondary) caries [1–3]. It is believed that areas of microleakage (gaps) between the tooth and composite are susceptible to secondary caries formation. Penetration of bacteria through the gaps causes pulp pathology and material degradation, which lead to clinical failure [4,5]. Microleakage of dental composites is often generated by polymerization shrinkage of methacrylate

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resins. Significant efforts have been focused on reducing the shrinkage of resin composites as well as developing methods to prevent microleakage by improving or optimizing the tooth-composite bonding.

Proper measurement techniques are necessary to correctly evaluate microleakage formation and propagation over time [6,7]. Current methods have been limited to sectioning the restored tooth, followed by imaging. The prevailing approach involves submerging a bonded sample of tooth and composite in a dye solution, such as an organic dye or silver nitrate solution, followed by vertically sectioning several consecutive slices. The depth of dye penetration from these sections is measured and reported as microleakage. This approach provides the correct penetration depth with uniformly distributed microleakage. However, more in-depth analyses have revealed that microleakage is seldom uniformly distributed [8]. Thus, the depth measured from the vertical sections generally fluctuates unpredictably as the gap measured for a selective area cannot represent the entire sample. Although labor-intensive and tedious, sectioning the tooth-composite structure from the top to the bottom and scanning each cross-section can provide a more accurate picture of the microleakage. However, it is not feasible to characterize a large number of samples using this more comprehensive sectioning approach. Furthermore, sectioning destroys the sample and renders additional testing impossible.

With the advent of advanced imaging/tomography techniques, the internal structures can be monitored nondestructively. Using these techniques, monitoring polymerization shrinkage and microleakage is possible if the precise structure at each location in space is known. Xray micro-computed tomography (µCT), which obtains the three-dimensional (3D) structures of small objects with high spatial resolution, has been widely accepted in biomedical research for examining the structure and mineral content of bones and teeth [9-12]. It has also been used to visualize structural features in tissue engineering scaffolds and subsequent tissue regeneration [13,14]. In dental materials research, µCT has been used to analyze the structures at dentin-adhesive-composite interfaces [15], to characterize the mineral content of enamel and dentin [16], and to evaluate 3D marginal adaptation qualitatively [17]. In the latter case, an accurate assessment of marginal adaptation depends on reliable volume calculations for the tooth and the restorative material. Results from µCT were also used to generate surface contours of enamel, dentin, and restorations, and analysis using 3D finite element analysis model [18]. In a previous study, we demonstrated that µCT can be utilized to accurately determine the volume of polymeric dental composites that have sufficient radiopacity. By comparing composite volumes before and after polymerization, the shrinkage in unconstrained containers can be obtained precisely and accu-

In the current study, we examine the use of  $\mu$ CT to determine shrinkage and microleakage in defined and constrained geometries with varied volumes and configuration factors (C-factors, defined as the ratio of contact to non-contact area). This is accomplished by tracking the volume and position changes of dental composites before and after photopolymerization. Rather than describing microleakage only in terms

of depth, we describe microleakage with more spatial information and with the aid of 3D visualization. Microleakage predictions obtained by  $\mu CT$  are also compared with results from a dye penetration method.

### 2. Materials and methods

#### 2.1. Materials<sup>1</sup>

Model cavities were prepared by assembling a polymethylmethacrylate (PMMA) tube with a defined inner diameter and a desired height and a PMMA rod (base) with the same outer diameter. For all samples, the cavity surface was sandblasted with  $50\,\mu m$  alumina for 15 s, followed by  $50\,\mu m$  silica for 15 s to create roughened surface for mechanical interlocking. Fiduciary markers (glass beads, diameter 125–150  $\mu m$ ) were embedded into the PMMA sample holders for sample position calibration. The resin composite used in this study was a commercially available hybrid composite (TPH, Dentsply Caulk, shade A1).

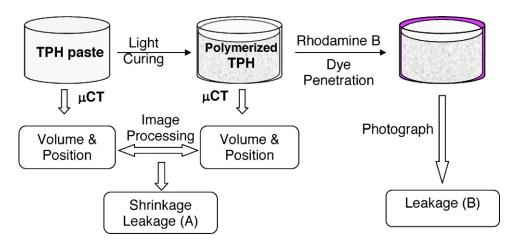
#### 2.2. X-ray micro-computed tomography

A Scanco Medical µCT 40 micro-computed tomography scanner was used to image the samples. The micro-focus X-ray source was set at 75 kVp and 114  $\mu A$  and the samples were scanned at a  $16\,\mu m$  line resolution with an integration time of 300 s. The composite pastes were packed into model cavities and fixed upright at marked positions in the µCT sample holder. The composites were then irradiated from 1 mm above the top of the composite for 60 s using a 600 mW cm<sup>-2</sup> Spectrum 200R (Dentsply Caulk) quartz tungsten halogen curing light equipped with an 8 mm 60° light guide. Another measurement was carried out 60 min after photopolymerization. 3D images were reconstructed and analyzed using the manufacturer's complete imaging and evaluation software and ImageJ image analysis software. The shrinkage was calculated using procedures described in our previous publication [19]. Calculated shrinkage is an average of six measurements. Three samples from each group were used to evaluate microleakage.

#### 2.3. ImageJ software

ImageJ software (version 1.39) was downloaded from the NIH website. 2D images of composites before and after polymerization obtained from  $\mu CT$  were imported into ImageJ. Image properties were verified and the dimensions of the images were set according to the voxel size of the  $\mu CT$  data. 2D images were stacked to form 3D images, which were used for calculation and further analysis. The 3D images generated for the sample after polymerization were subtracted from the images generated before polymerization. The volume difference

<sup>&</sup>lt;sup>1</sup> Certain equipment, instruments or materials are identified in this paper to adequately specify the experimental details. Such identification does not imply recommendation by the National Institute of Standards and Technology or the American Dental Association, nor does it imply the materials are necessarily the best available for the purpose.



Scheme 1 – Illustration of experimental design. The composites in model cavities were fixed to sample holders and measured before and after photopolymerization. The polymerization shrinkage was determined by comparing the volume and position of composites before and after polymerization. The possible leaking areas/regions were predicted after analysis. The same sample was subsequently immersed in a Rhodamine B dye solution. Dye penetration provided visual assessment of leakage position as supplemental information to compare to microleakage results derived from  $\mu$ CT.

between the image stacks predicted possible regions of leakage and was illustrated using 3D projection.

# 2.4. Dye penetration

After the post-polymerization  $\mu CT$  measurements, samples were immersed into an aqueous Rhodamine B solution (1% by mass) for 24h. The pink dye penetrated into the composite/holder interface immediately. Dye penetration was captured for direct visualization.

#### 2.5. FTIR microspectroscopy

FTIR microspectroscopy analyses were performed using a Nicolet Magna-IR 550 FTIR spectrophotometer interfaced with a Nic-Plan IR microscope operated in reflectance mode (FTIR-RM). The microscope was equipped with a video camera, a liquid nitrogen-cooled mercury cadmium telluride (MCT) detector and a computer-controlled x-y translation stage. A total of 64 scans were collected from 650 cm<sup>-1</sup> to 4000 cm<sup>-1</sup> at  $8 \, \text{cm}^{-1}$  resolution with a beam spot size of  $90 \, \mu\text{m} \times 90 \, \mu\text{m}$ . Five spectra each were obtained from the flat top and bottom of each filled cavity (from their middle and from 0.5 mm to their right, left, above and below). Each spot was manually focused before data collection. The reflectance spectra were proportioned against a background of an aluminum mirror and transformed to absorbance spectra using the Kramers-Kronig transform algorithm for dispersion correction, which converts the reflectance spectra to absorbance-like spectra [20,21]. The degree of conversion (DC) was calculated as the reduction in the methacrylate peak (1634 cm<sup>-1</sup>) height using the phenyl absorbance peak  $(1610 \, \text{cm}^{-1})$  as an internal standard.

Composite-filled PMMA tubes were sandwiched between two glass cover slips in order to obtain the smooth top and bottom surfaces that are required for FTIR-RM. During photopolymerization, samples were placed onto a black surface to reduce light reflection, and the curing light radiated from 1 mm above the top surface of the samples. Each sample was cured for 60 s and the FTIR-RM spectra were obtained from both the top (close to curing light) and bottom surfaces at 1 h and 24 h after light curing. The DC was calculated. The results are an average of five measurements.

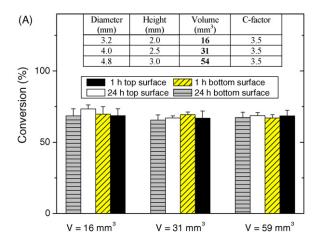
#### 2.6. Statistical analysis

Polymerization shrinkage values determined using  $\mu$ CT and DC measured by FTIR-RM were analyzed using one-way analysis of variance (ANOVA) with a 95% confidence interval to indicate significant differences.

# 3. Results and discussion

The general procedures for determining the polymerization shrinkage and corresponding microleakage are illustrated in Scheme 1.  $\mu$ CT was used to measure the composite volumes before and after photopolymerization and shrinkage was calculated based on the percentage of volume change upon polymerization. The current study examines the effect of C-factor and sample volume on polymerization shrinkage and the corresponding microleakage, which was determined by assessing the distribution of polymerization shrinkage at the composite–cavity interface. A parallel experiment to characterize microleakage was carried out by dye penetration experiments to validate the  $\mu$ CT results.

To obtain a meaningful comparison of shrinkage and leakage results when evaluating composites placed in cavities of differing C-factors and volumes, it is imperative for all samples to have an identical DC; therefore, DC of composites was examined using FTIR-RM [22]. Fig. 1 illustrates the DC of composites at the top and bottom surfaces 1h and 24h post-cure. All samples reached approximately 69% conversion regardless of position and time after irradiation. The statistical analysis of variance indicates that there was no significant difference among samples that have the same C-factor but



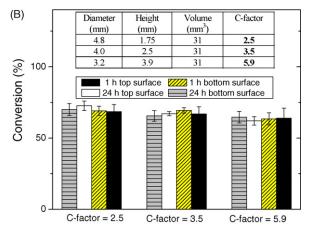


Fig. 1 – Degree of conversion of composites measured using FTIR-RM. (A) DC of composite with the same C-factor but different volumes and (B) DC of composites with the same volume but different C-factors. Results are an average of five measurements.

different volumes (Fig. 1A) or have the same volume but different C-factors (Fig. 1B). The same DC at the top and bottom of the samples suggests that the light irradiation was equally effective at all positions and polymerization kinetics were relatively uniform. The transparency of the PMMA sample holders allowed light to penetrate as far as 3.9 mm to produce evenly polymerized materials. These results confirm that the shrinkage and leakage results obtained by  $\mu CT$  in different model cavities can be compared directly.

The composite shrinkage as a function of C-factor and sample volume was examined using  $\mu$ CT (Fig. 2). In previous studies, different shrinkage volumes were often observed for the same material using different test methods [23]. These variations in polymerization shrinkage have been attributed in part to differences in the test set-up, generally associated with the C-factor and sample volume. The current study separates the two parameters, holding one constant and examining the effect of each parameter independently. The shrinkage of the composite ranged between 2.6% and 2.9% for all configurations and the statistical analysis showed no significant differences (p-value <0.05) in shrinkage among samples with different C-factors or volumes. These results

indicate that neither C-factor nor sample volume affects polymerization shrinkage at a constant DC. Consequently, we attribute the observed differences in interfacial volume changes and subsequent microleakage to the sample geometry.

In addition to measuring the shrinkage, the same samples were used to assess positional changes in order to predict sites for potential microleakage. One critical issue for achievement of reliable results is time-resolve image alignment. In other words, a direct data comparison of pre- and post-polymerization images requires the sample position to be stationary. Since minor position shifts can occur upon scanning, fiduciary markers were inserted into sample holders to correct for any sample shifts during the measurement. The centers of mass of fiduciary markers were determined and noted as reference positions. Images from before and after polymerization were shifted in the three principal axes such that the reference positions matched precisely.

We first evaluate shrinkage as a function of sample depth using the following method. For each slice, we calculate an area difference ( $\Delta A$ ) of the composite before and after photopolymerization (Fig. 3A, inset), from which the volume difference ( $\Delta V$ ) was determined ( $\Delta V = \Delta A \times h$ , where h is the height of each slice, 16 µm). The volume difference is plotted as a function of sample depth (Fig. 3). For all samples, the majority of the shrinkage occurred at the top and bottom of the composites, but shrinkage was also observed in the mid-section of the samples, indicating sites for potential leakage. When comparing samples with the same volume but different C-factors, composites with the largest C-factor (5.9) exhibited the highest shrinkage in the mid-section (Fig. 3A), whereas composites in holders with lower C-factors showed a smaller volume change in the bulk of the sample. The percentage of total shrinkage in the mid-section (approximately 54% of the total height) was 9.3%, 17.2%, and 43.6% of total volume change for C-factors of 2.5, 3.5, and 5.9, respectively. Since overall shrinkage is the same for all samples and occurs predominately at the top, geometries with lower C-factors

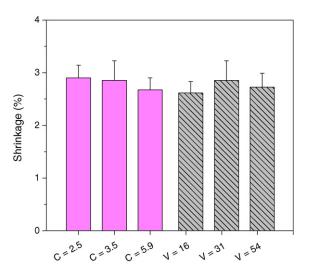


Fig. 2 – Polymerization shrinkage (n = 6) determined for the same composite under constrained sample geometries as a function of C-factor and sample volume (mm<sup>3</sup>) using  $\mu$ CT.

can accommodate a greater shrinkage volume at the free surfaces and thus reduce shrinkage distributed throughout the constrained sample surfaces. For samples with the same C-factor (3.5) but different volumes, shrinkage was directly correlated with sample volume (Fig. 3B). The percentage of volume changes of the mid-section increased as the volume increased, i.e., 12.1%, 17.2%, and 24.3% of total volume changes for cavity volume of 16 mm<sup>3</sup>, 31 mm<sup>3</sup>, and 54 mm<sup>3</sup>, respectively. These results are expected since a larger composite volume results in a greater overall volume change, which should lead to greater stress and leakage. Another way to correlate microleakage with volume change is to calculate a gap (distance between delaminated composite and cavity wall) using the volume difference between cured and uncured composite as a function of depth (gap=volume/area of the cavity-composite interface). The calculated gaps in the midsection of samples listed in Fig. 3A and B are shown in Fig. 3C

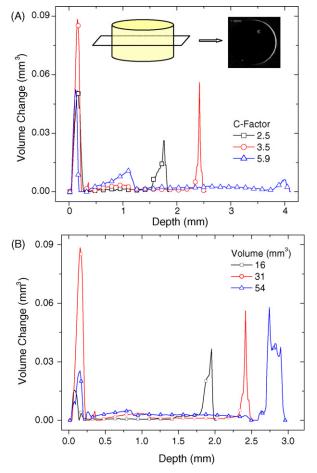
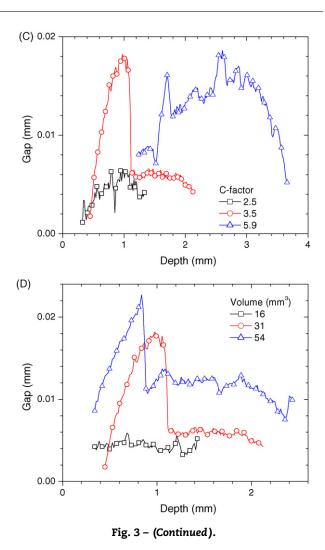


Fig. 3 – Volume changes measured as a function of sample depth (at  $16\,\mu m$  intervals): (A) a series of samples with constant volume and varied C-factors, inset shows the area difference between polymerized and unpolymerized composite for one slice, from which the volume difference was calculated and plotted as one data point along the sample depth, and (B) a series of samples with constant C-factor and varied sample volumes; gaps in the central portion as a function of depth: (C) a series of samples with constant volume but varied C-factors, and (D) a series of samples with constant C-factor but varied sample volumes.



and D, respectively, and confirm that a larger C-factor or volume results in a larger gap.

The data illustrated in Fig. 3 represent volume and average gaps at each slice. This is inadequate to predict the spatial distribution of microleakage. In the following, we use two analysis approaches to quantify the spatial distribution of microleakage and compare those results with images captured from dye penetration experiments. The first approach involves further sectioning the data (described in Fig. 3)—each slice is divided into 12 equally spaced  $30^{\circ}$  arcs. The depth of the sample is analogous to latitude and the arcs are analogous to longitude on a map. The gaps are calculated as previously described but for a smaller area (1/12 of the unit area). A location is designated as "leaking" if the gap at a location is greater than 16 µm; likewise, designated as "likely leaking" if the gap is between  $0 \mu m$  and  $16 \mu m$ ; and designated as "not leaking" if no gap is found. Based on these criteria, a microleakage map can be generated and presented in 2D. An example is shown in Fig. 4A (C-factor = 3.5 and volume = 54 mm<sup>3</sup>). The leakage map clearly indicates that the microleakage was highly nonuniform. For all the samples shown in Fig. 3, the analysis showed that one side of the composites delaminated from the sample holder while the other side remained in contact.

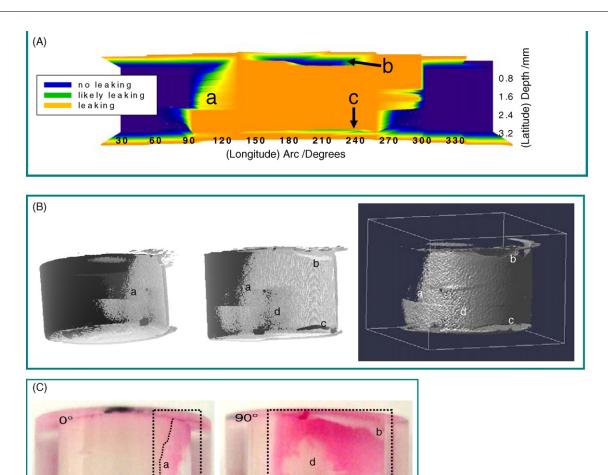


Fig. 4 – (A) 2D leakage map determined by  $\mu$ CT evaluation where the circumference is plotted on the x-axis and the sample depth is plotted on the y-axis (refer to text for the definition of leaking); (B) the left two images are 3D leakage maps from different views where the leakage is in light gray and the composite is in dark gray; the right image is a 3D side view of the leakage only; the middle and right images share the same view point; and (C) optical images of the same sample captured between 0° and 90°, corresponding to angles are shown in Fig. 4A. The leakage regions are enclosed in dotted lines. Letters a, b, c, and d point to four identical locations for both image processing and dye penetration results.

The second analysis approach compares the composite before and after polymerization voxel-by-voxel. The analysis was performed with the aid of an image analysis macro written for ImageJ software that can rotate and translate the  $\mu$ CT reconstructions in 3D space. This capability allows for the comparison and calculation of pre- and post-polymerization 3D reconstructions, thus providing a 3D visualization for the microleakage. Fig. 4B illustrates three projection views of microleakage for the same sample from different viewing angles. The uneven distribution of microleakage can be clearly visualized.

The 2D and 3D leakage maps were compared with photographs of the same sample captured at different angles after dye penetration (Fig. 4C). The same features are labeled as "a–d" for all 2D and 3D leakage maps, where "a" describes the curvature of leakage region at approximately 90°; "b" is a void region as indicated by the white color in Fig. 4B (middle image), but can be misinterpreted as no leakage from other data presentations; "c" is a region with no leakage; "d" denotes a complex region where a reticular leakage pattern consisted

of discontinuous islands is expected from the dye penetration and pixel-by-pixel calculation, similar to previously observed nanoleakage structures [24]. However, more crude 2D leakage map cannot provide the leakage pattern and simply indicate this region as non-leaking. In general, the leakage information predicted by  $\mu CT$  agreed well with results from dye penetration in terms of the depth and overall spatial distribution. These results highlight the complexity of leakage patterns and the advantages of 3D leakage mapping, which better correlate with dye penetration (true) leakage results. The composite with the largest C-factor and the composite with the largest volume had complete dye penetration from the top to the bottom. All other samples showed appreciable dye penetration.

Each of the two  $\mu$ CT analysis approaches has its advantages and deficiencies. The first approach that uses sectioning followed by gap calculation provides a reasonable spatial distribution, although an improved spatial resolution is attainable if the sample is sectioned into more arcs. The gap values calculated based on a sufficiently large number of data points can be used to reasonably predict the magnitude of large gaps.

The second approach that involves voxel-to-voxel comparison gives good spatial resolution and provides a visualization tool. However, the results may be more qualitative in term of the gap dimension as the magnitude of any predicted leakage cannot be accurately evaluated. Dye penetration experiments provide spatial resolution in model PMMA cavities, but provide relatively little information in actual teeth without laborious sectioning and imaging.  $\mu$ CT analyses can provide an equal amount of information without destroying the sample. The microleakage maps presented in this work are an improvement over previous quantification methods that typically present microleakage only in terms of depth.

#### 4. Conclusions

X-ray micro-computed tomography is effective in determining the precise volume and position of polymeric dental composites in model cavities, from which polymerization shrinkage and microleakage may be determined. At the same DC, polymerization shrinkage results in model cavities of different C-factors and volumes were statistically identical. In all model cavities, the composite contracted predominantly from the top and bottom surfaces, although shrinkage was found throughout the entire depth of the sample. Microleakage characterized as gaps between the cavity wall and the composite was highly dependent on C-factor as well as composite volume, with higher C-factors and larger volumes leading to greater probability of microleakage. Microleakage results predicted by  $\mu$ CT analysis combined with 3D image analysis agreed well with those obtained by dye penetration. For all samples evaluated, the microleakage was not uniform, highlighting the need for a comprehensive evaluation. The primary advantages of the methods presented in this study include the ability to characterize microleakage regardless of sample geometry and the non-destructive nature of these measurements, which permit further evaluation on intact samples.

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